

FORMATION OF METASTABLE VATERITE CRYSTALS IN A POLYMER MATRIX BY GASEOUS DIFFUSION

MICHAEL W. PITCHER^{A*}, ABDOLLAH NESHAT^B AND SEMIH SEYYIDOĞLU^A

^aDepartment of Chemistry, Middle East Technical University, 06531 Ankara, Turkey

^bCurrent address: Department of Chemistry, Bowman-Oddy Laboratories, MS 602, The University of Toledo, 2801 W. Bancroft St, Toledo, OH 43606, USA

(Received: 24 May 2007 - Accepted: 27 February 2008)

ABSTRACT

A composite consisting of a polymer and calcium carbonate has been synthesized by allowing carbon dioxide to diffuse into a poly(ethylene oxide) film containing a source of calcium ions. The X-ray pattern of the synthetic composite material shows that the least thermodynamically stable polymorph of calcium carbonate, vaterite is the mineral phase which is formed, rather than the most thermodynamically stable calcite phase. This effect has been seen in other reports of crystallization of inorganic phases which have been mediated by a polymer matrix and/or other organic moieties. In this paper, some possible explanations are presented which may help explain this phenomenon.

INTRODUCTION

The crystallization behavior of calcium carbonate (CaCO_3) is important for multiple reasons. Although it is not the most technologically important material, CaCO_3 is an unwanted problem in many industries, including the paper industry¹ and in water treatment². Recent work is also showing that it can indeed be a beneficial component in the formation of polymer³ and plastic/rubber⁴ CaCO_3 nanocomposites with improved properties. Perhaps the most important reason is trying to understand how living organisms control the crystallization of CaCO_3 in various skeletal parts; such an understanding might provide access to new synthetic strategies and novel or functional materials, as well providing insight into the biomineralization process^{5,6}. One approach to mimicking the natural biomineralization process and/or attempting to control or affect the crystallization behavior of CaCO_3 is to carry out the synthesis in the presence of a polymer or polymeric additive, which may or may not be biological in origin. In the last year nearly one hundred English journal papers have been published on this topic alone; in ten of these cases⁷⁻¹⁶, vaterite, the least thermodynamically stable polymorph of CaCO_3 is formed, rather than the most thermodynamically stable form, calcite. This is analogous to the phase selection seen in biomineralized sea-shells and other structural materials.

In this paper we describe a new method for synthesizing vaterite, by allowing carbon dioxide (CO_2) to diffuse into a polymer film containing a source of calcium ions (calcium chloride) and discuss possible reasons why the metastable vaterite is produced instead of calcite. The polymer chosen was poly(ethylene oxide) as it has been previously been demonstrated that there is strong interaction between this polymer and growing inorganic salts causing control of phase and/or morphology^{17,18}.

EXPERIMENTAL

Poly(ethylene oxide) (PEO, $M_w=200,000$) and calcium chloride ($\text{CaCl}_2 \cdot 2\text{H}_2\text{O}$) were purchased from Aldrich and used as received. Composite CaCO_3/PEO films were prepared by casting a film of PEO/ CaCl_2 from solutions of PEO and $\text{CaCl}_2 \cdot 2\text{H}_2\text{O}$ (9:1 ratio by weight to yield a total weight of 0.50 g) dissolved in 10 mL distilled water. The solution was poured onto a glass microscope slide until no more solution could be applied. The solvent was then allowed to evaporate. The film was removed from the glass slide prior to the reaction using a razor blade or scalpel. The thickness of the film was between 0.1 and 0.2 mm. The film was then placed in dessicator and suspended above a beaker containing solid ammonium carbonate ($(\text{NH}_4)_2\text{CO}_3$) as the source of CO_2 . After twenty four hours the films were removed and analyzed by optical microscopy and X-ray diffraction (XRD). In addition to this a beaker of CaCl_2 solution was also placed in the dessicator with $(\text{NH}_4)_2\text{CO}_3$, the white precipitate that begins to form almost immediately was filtered, dried and the X-ray pattern also taken. This experiment acted as a control.

X-ray analysis of the compounds was performed using a Rigaku Diffractometer Miniflex equipped with $\text{CuK}\alpha$ (30 kV, 15 mA, $\lambda=1.54051 \text{ \AA}$) radiation at room temperature. Data was collected in continuous scanning mode with steps of 0.05° (2θ) with an angular range from $2\theta=10$ to 70° . The divergence slit was variable, and the scattering and receiving slits were 4.2 deg. and 0.3 mm, respectively. X-ray patterns were obtained of the starting $\text{CaCl}_2 \cdot 2\text{H}_2\text{O}$ powder, a pure PEO film, a PEO/ CaCl_2 film, the final PEO/ CaCO_3 film and the white precipitate from the control experiment. Powder

X-ray diffraction (XRD) data of the control experiment was collected on a PANalytical X'Pert Pro using $\text{CuK}\alpha$ radiation again between $2\theta=10$ to 70° . The sample was contained in an aluminum sample holder. Optical microscopy was done using a Leica CME.

RESULTS AND DISCUSSION

The X-ray patterns of the samples are presented in Fig. 1. It should be noted that there is no evidence of crystallization on the polymer surface, so crystallization has to have occurred within the polymer film; this was deduced by optical microscopy studies.

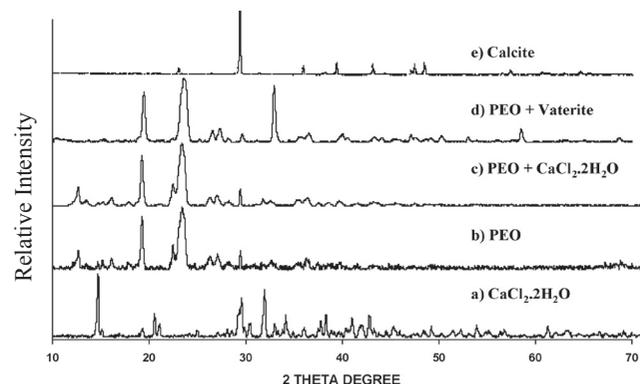


Fig. 1. X-ray powder diffraction patterns of a) calcium chloride starting material, b) a poly(ethylene oxide) film, c) a poly(ethylene oxide) and calcium chloride film, d) CaCO_3 (vaterite) grown in a poly(ethylene oxide) film, e) CaCO_3 (calcite) precipitated by exposure of a calcium chloride solution to a source of carbon dioxide.

By comparing patterns a with b and c, it is clear that the PEO reflections dominate the pattern of the PEO/ CaCl_2 starting material. Upon subsequent exposure to CO_2 however there is a distinct change in the pattern (d). Analysis of pattern d is presented in Table 1.

Table 1. X-ray powder diffraction analysis of pattern d).

Pattern d	Vaterite		PEO			
d_{obs}	I/I_0	d_{obs}	I/I_0	d_{obs}	I/I_0	Identification
4.5715	78			4.6187	87	PEO
3.9744	6			3.9656	41	PEO
3.7824	100	3.713	8	3.8144	100	PEO
3.5379	4	3.576	42			Vaterite
3.3606	19			3.3921	23	PEO
3.2698	22	3.294	74			Vaterite
2.8687	3			2.8959	8	PEO
2.7200	88	2.733	100			Vaterite
2.4629	14			2.4793	18	PEO
2.2548	12	2.279	2	2.2878	8	PEO
2.2281	7	2.218	17			Vaterite
2.0878	8			2.0786	8	PEO
2.0517	7	2.064	50			Vaterite
1.8521	6	1.856	32			Vaterite
1.8158	10	1.824	41			Vaterite
1.7278	9	1.749	10			Vaterite
1.5776	22	1.569	1			Vaterite
1.3668	7	1.367	9			Vaterite

The figure shows that, again, the pattern is dominated by the PEO reflections, however there are new reflections, which are neither the CaCl₂ starting material nor calcite (control – pattern e (note: the small peaks at $2\theta = 38, 45$ and 65 belong to the aluminum sample holder)) but which were determined to be vaterite (ICDD Card No: 74-1867) (Table 1).

The question is why is the metastable vaterite formed in the presence of the polymer, but calcite in the absence of PEO? There are reports where crystallizing an inorganic phase inside a polymer has produced high-pressure phases of cadmium sulfide¹⁹, novel and high-pressure phases of lead sulfide²⁰ and in one instance the formation of crystalline lithium niobate without the application of heat²¹. It is suggested that “the density or rigidity of the matrix regulates the phase of the crystals that form, predominantly by a “solid state” densification effect, whereby the solid state of the polymer matrix acts to confine growth spatially such that the growing crystal is forced to adopt a more dense form than it would in solution growth, resulting in the formation of high-pressure phases”²².

In this work, however, this is unlikely to be the explanation for vaterite formation, for two reasons. Firstly, vaterite is not the most dense form of CaCO₃ (aragonite > calcite > vaterite) and secondly, vaterite is not, to the best of our knowledge, the high-pressure phase of CaCO₃; although to date, we have only been able to find one phase diagram of CaCO₃²³ in existence. So an alternative explanation has to be found; one theory that is often seen in literature reports when vaterite is formed is that there is some kind of stabilization of the vaterite phase which stops it transforming to calcite, or growth inhibition of the stable calcite²⁴. This theory works very well but still does not explain why vaterite is formed in the first instance. A possible, perhaps more convincing argument is that there is some effect on the surface energy of the growing crystal, which affects the phase into which it first crystallizes. The pioneering work of Navrotsky and co-workers has shown that the different polymorphs of various inorganic minerals, including alumina²⁵, titania²⁶ and zirconia²⁷ have different surface energies, which in turn is a key contributing element to the stability of that form. Also the elegant work by De Yoreo, Orme and co-workers has shown that, specifically in the case of CaCO₃, the binding of amino acids (the building block of biological polymers; polypeptides) changes the free energies of the step edges of calcite crystal thus altering “the energy landscape”²⁸. To prove or disprove such a theory is beyond the scope of this work, but one aspect remains surely certain and that is that the energetics of crystal growth in a polymer are likely to be very different than crystal growth by precipitation or other routes. Our studies continue in this area by looking at the stability of the vaterite formed in the polymer films and attempting to follow the crystallization/crystal growth by transmission electron microscopy.

CONCLUSIONS

Vaterite, the least thermodynamically stable form of CaCO₃ has been formed by a new method carrying out crystallization within a polymer film. This formation of a metastable or thermodynamically unfavorable phase has been seen by other workers with a variety of different inorganic minerals; possibly indicating that such a methodology might be used for the synthesis of other technologically important materials e.g. cubic boron nitride under similarly mild conditions.

ACKNOWLEDGEMENTS

The authors would like to thank John Neil and Sergey Ushakov of the Thermochemistry Facility at the University of California, Davis for help in retrieving reference 23 and Cora Lind at the University of Toledo for obtaining the X-ray pattern of the control experiment (calcite).

REFERENCES

1. A. Loraine, P.E. Huchler, *Paper Technology* **47**, 23, (2006).
2. D. Hasson, R. Semiat, M. Ilevicky, D. Damiano, A. Sher, *Proceedings - Water Quality Technology Conference and Exhibition* (2004).
3. X. Lu, T. Liu, *Polymer Nanocomposites* **412**, (2006).
4. A. Chakravarty, *Rubber India* **57**, 23, (2005).
5. S. Mann *Biomaterialization: Principles and Concepts in Bioinorganic Materials Chemistry*, Oxford University Press, Oxford, 2002.
6. A-W. Xu, Y. Ma, H. Cölfen, *J. Mater. Chem.* **17**, 415, (2007).
7. K. Wikander, P. Kjellin, K. Holmberg, *Nord. Pulp Paper Res. J.* **21**, 286, (2006).
8. A. Sugawara, A. Oichi, H. Suzuki, Y. Shigesato, T. Kogure, T. Kato, *J. Polym. Sci. Polym. Chem.* **44**, 5153, (2006).
9. K. Naka, S-C. Huang, Y. Chujo, *Langmuir* **22**, 7760, (2006).
10. X-H. Guo, S-H. Yu, G-B. Cai, *Angew. Chem., Int. Ed. Engl.* **45**, 3977, (2006).
11. Y-X. Gao, S-H. Yu, X-H. Guo, *Langmuir* **22**, 6125, (2006).
12. A. Kotachi, T. Miura, H. Imai, *Cryst. Growth Des.* **6**, 1636, (2006).
13. K. Ichikawa, N. Shimomura, B. *Chem. Soc. Jpn* **79**, 580, (2006).
14. N. Loges, K. Graf, L. Nasdala, W. Tremel, *Langmuir* **22**, 3073, (2006).
15. H. Wakayama, S.R. Hall, Y. Fukushima, S. Mann, *Ind. Eng. Chem. Res.* **45**, 3332, (2006).
16. H. Shinya, O. Kousaku, Y. Hiroyuki, *Macromol. Biosci.* **6**, 228, (2006).
17. S. Radhakrishnan, J.M. Schultz, *J. Cryst. Growth* **116**, 378, (1992).
18. S. Radhakrishnan, D.R. Saini, *J. Cryst. Growth* **129**, 191, (1993).
19. J. Lin, E. Cates, P.A. Bianconi, *J. Am. Chem. Soc.* **116**, 4738, (1994).
20. M.W. Pitcher, E. Cates, L. Raboin, P.A. Bianconi, *Chem. Mater.* **12**, 1738 (2000).
21. M.W. Pitcher, Y. He, P.A. Bianconi, *Mater. Chem. Phys.* **90**, 57, (2005).
22. E. Cates, M.W. Pitcher, P.A. Bianconi, *Mater. Chem. Phys.* **94**, 13, (2005).
23. A.F. Bol'shakov, L.V. Abramova, B.V. Abalduev, *J. Appl. Chem. USSR (Engl. Transl.)* **49**, 2076, (1976).
24. I.W. Kim, R.E. Robertson, R. Zand, *Cryst. Growth Des.* **5**, 513, (2005).
25. J.M. McHale, A. Auroux, A.J. Perrotta, A. Navrotsky, *Science* **277**, 788, (1997).
26. M. R. Ranade, A. Navrotsky, H. Z. Zhang, J. F. Banfield, S. H. Elder, A. Zaban, P. H. Borse, S. K. Kulkarni, G. S. Doran, H. J. Whitfield, *P. Natl Acad. Sci. USA* **99**, 6476, (2002).
27. M.W. Pitcher, S.V. Ushakov, A. Navrotsky, B.F. Woodfield, G. Li, J. Boerio-Goates, B.M. Tissue, *J. Amer. Ceram. Soc.* **88**, 160 (2005).
28. C.A. Orme, A. Noy, A. Wierzbicki, M.T. McBride, M. Grantham, H.H. Teng, P.M. Dove, J.J. DeYoreo, *Nature* **411**, 775, (2001).